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6 **Gallium-modified chitosan/poly(acrylic acid) bilayer coatings for improved**
7 **titanium implant performances**

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27

28 **Abstract**

29 A gallium-modified chitosan/poly(acrylic acid) bilayer was obtained by electrochemical techniques
30 on titanium to reduce orthopaedic and/or dental implants failure. The bilayer *in vitro* antibacterial
31 properties and biocompatibility were evaluated against *Escherichia coli* and *Pseudomonas*
32 *aeruginosa* and on MG63 osteoblast-like cells, respectively. Gallium loading into the bilayer was
33 carefully tuned by the electrochemical deposition time to ensure the best balance between
34 antibacterial activity and cytocompatibility. The 30 minutes deposition time was able to reduce *in*
35 *vitro* the viable cell counts of *E. coli* and *P. aeruginosa* of 2 and 3 log cfu/sheet, respectively. Our
36 results evidenced that the developed antibacterial coating did not considerably alter the mechanical
37 flexural properties of titanium substrates and, in addition, influenced positively MG63 adhesion and
38 proliferation. Therefore, the gallium-modified chitosan/poly(acrylic acid) bilayer can be exploited
39 as a promising titanium coating to limit bacterial adhesion and proliferation, while maintaining
40 osseointegrative potential.

41

42 **Keywords:** Chitosan-based Bilayer, Gallium, Titanium, Antibacterial agents, Electrochemical
43 deposition, Biocompatibility.

44

45 **Chemical compounds studied in this article**

46 Chitosan hydrochloride (PubChem SID: 295370713); gallium (III) nitrate hydrate (PubChem CID:
47 11550823); poly(acrylic acid) (PubChem CID: 6581); titanium (PubChem CID: 23963); ammonium
48 peroxydisulfate (PubChem CID: 62648).

49 **1. Introduction**

50 Large bone defects are not a rare issue to deal with, and when high load-bearing implants are
51 required, Titanium (Ti) and its alloys represent the gold standard. Among the noteworthy features of
52 Ti, there is the match between lightweight and desirable Young's modulus, as well as the durability
53 given by corrosion resistance (Rack & Qazi, 2006). Since the implant surface is in direct contact
54 with the surrounding tissues, several strategies have been promoted in order to enhance Ti implants
55 biocompatibility (Sidambe, 2014) and, among these, the development of bioactive polymeric
56 coatings represents one of the most effective approaches.

57 Microbial colonization and biofilm formation are among the major causes of orthopaedic and/or
58 dental titanium-based implants failure (Hedrick et al., 2006). The local antimicrobial delivery
59 represents a well-experimented smart strategy to hinder bacterial adhesion and proliferation,
60 avoiding the side effects arising from a massive systemic antibiotic administration (Trampuz et al.,
61 2003). In this respect, the implant coating could be also designed in order to perform a main role in
62 infection prevention.

63 Gallium (Ga) has shown bacteriostatic activity and efficacy in the treatment of several disorders
64 associated with bone resorption, including Paget's disease and cancer-related hypercalcemia
65 (Bernstein, 1998; Bernstein, 2013). Furthermore, it potentiates the antimicrobial action of
66 lactoferrin (Berlutti et al., 2008) and stops bone resorption by quenching osteoclasts activity
67 (Verron et al., 2010). Moreover, gallium has recently shown efficacy in infections treatment:
68 exploiting the chemical similarity with Fe^{3+} , Ga^{3+} can interfere with iron metabolism of a wide
69 range of bacteria (Kelson et al., 2013).

70 Poly(acrylic acid) (PAA), is stable, non-toxic, non-inflammatory and able to mimic bone-
71 surrounding tissues. PAA coatings, synthesized on Ti or its alloys via an electrochemical process,
72 have been previously studied, highlighting both biocompatible and anticorrosion performances for
73 metallic implants (De Giglio et al., 2007). In this work, poly(acrylic acid) has the role to promote
74 the adhesion of a gallium-modified chitosan layer on titanium.

75 Chitosan (CS) is a biopolymer that shows biocompatible, biodegradable, osteoconductive, and wound
76 healing properties (Muzzarelli et al. 2009), thanks to its structural and chemical homology to
77 proteoglycans found in the extracellular matrix. For all these reasons and also thanks to its ability to
78 complex metal ions (Varma et al., 2004), it was selected as vehicle to deliver the metallic antibacterial
79 agent to the local implant-tissue interface.

80 In this work, a two-step electrochemical procedure was used to develop on titanium a polymeric
81 bilayer, obtained by electropolymerization of a PAA layer followed by electrochemical deposition
82 of CS-Ga layer, with the double aim of preventing infections and enhancing implant
83 biocompatibility.

84 An accurate physico-chemical, morphological and mechanical characterizations of PAA/CS-Ga
85 bilayer were reported. Antibacterial performances against *E. coli* and *P. aeruginosa* were evaluated,
86 as well as compatibility with MG63 human osteoblast-like cells.

87

88 **2. Materials and methods**

89 **2.1 Materials**

90 All reagents purchased from commercial sources were used as provided unless otherwise stated.

91 Acrylic acid (AA), used as monomer, ammonium peroxydisulfate ((NH₄)₂S₂O₈), used as
92 electrochemical initiator, and gallium nitrate (Ga(NO₃)₃), used as source of gallium ions, were
93 purchased from Sigma Aldrich® (Milan, Italy). Chitosan hydrochloride (deacetylation degree
94 86.4%, viscosity 4 mPa·s, approx. molecular weight 30-400kDa, according to manufacturer) was
95 obtained from HEPPE MEDICAL CHITOSAN GmbH (Germany). Titanium sheet electrodes were
96 mechanically polished by fine diamond paper and then by Al₂O₃ powder (50 μm). After this
97 treatment, before polymer deposition, each electrode (2 cm²) was cleaned by an ultrasonic bath
98 using ethanol and successively triple-distilled water. All microbiological media and supplements
99 were purchased from Biolife Italiana srl (Milan, Italy). The materials used to determine the gallium

100 amount loaded and released by the coating using DPASV were reported in the Supplementary data
101 (S1.1 Paragraph).

102

103 **2.2 Bilayer electrosynthesis**

104 All electrochemical experiments were carried out using a PAR VersaSTAT4 potentiostat-
105 galvanostat (Princeton Applied Research, Hampshire, UK) and a three-electrode cell. A titanium
106 sheet was used as cathodic working electrode, while a platinum wire was the anodic one.

107 All potentials were related to the reference system used: Ag/AgCl (KCl sat.) in water (0.197V vs
108 SHE at 25°C). PAA coatings were prepared directly on titanium by cyclic voltammetry in aqueous
109 solution of acrylic acid (AA) 0.1M and ((NH₄)₂S₂O₈) 0.1M (experimental parameters: initial
110 potential 0.0V; final potential -1.2V; scan rate 100mV/s; 20 cycles) (De Giglio et al., 2007).

111 Successively, CS and Ga ions were electrodeposited on the coating by chronoamperometry (-1.5V).
112 Three different deposition times (15, 30 and 60 minutes) were used to obtain the CS-Ga layer over
113 PAA. The dried samples were sterilized by UV treatment before *in vitro* antibacterial and
114 cytocompatibility tests.

115

116 **2.3 Gallium quantification and release**

117 The gallium amount loaded in or released by the bilayer was evaluated, after acid digestion, by
118 differential pulse anodic stripping voltammetry (DPASV). The same technique was previously used
119 by other authors to determine gallium traces (Udisti & Piccardi, 1988). However, in this work the
120 electrolytic composition of the sample solution and the instrumental settings were carefully
121 modified in order to optimize the measurement conditions.

122 Gallium release evaluation was performed, under stirring, dipping coated titanium sheets in 1mL of
123 PBS (pH 7.4) at 37°C for 12, 24, 48, 72, 120 and 168 h. After the incubation time, the PBS was
124 subjected to acid digestion and the gallium released quantified by DPASV.

125 The acid digestion was carried out according to a previously reported procedure (Danielsson et al.,
126 1981). The digested samples were then cooled at room temperature and diluted with distilled water
127 to a final volume of 25 mL. All gallium measurements were performed on a 757VA Computrace
128 voltammetric analyzer (Metrohm srl, Origgio, VA, Italy). Details on the cell setup, calibration and
129 measurements were reported in the Supplementary data (S1.1. Paragraph).

130

131 **2.4 Physico-chemical, morphological and mechanical characterization**

132 *2.4.1 Fourier Transform Infrared Spectroscopy in Attenuated Total Reflection mode (FT-IR/ATR)*

133 FT-IR/ATR analysis was performed on a Spectrum Two PE instrument using the Universal ATR
134 accessory (Single Reflection Diamond) (PerkinElmer Inc, Waltham, MA) at 4 cm⁻¹ resolution. The
135 samples were analysed without any preliminary preparative step.

136

137 *2.4.2 X-ray Photoelectron Spectroscopy (XPS)*

138 XPS was performed by a Thermo VG Thetaprobe spectrometer (Thermo Fisher Scientific, Inc.,
139 Waltham, MA, USA), equipped with a microspot monochromatized AlK α source. Survey scans
140 (binding energy (BE) range 0–1200 eV, FAT mode, pass energy 150 eV) and high-resolution spectra
141 (FAT mode, pass energy 50 eV) were recorded for each sample. Data analysis of the latter was
142 performed using the Avantage software package (version 5.937), which consists of a non-linear least
143 squares fitting program. The experimental points of the detailed spectra were fitted using Gaussian–
144 Lorentzian peaks having the same full width at half maximum (FWHM), according to fitting models
145 applied. Charge referencing was performed by setting the lower binding energy C1s photo-peak (i.e.,
146 C1s hydrocarbon peak) at 285.0 eV. Quantification (atomic percentage, At %) was made using
147 normalized peak area. The normalization of the peak area and comparison of data from different
148 element was enabled by correction with empirically derived sensitivity factors according to Scofield
149 libraries (Scofield, 1976) and using the formula already reported (De Giglio et al., 2013).

150

151 *2.4.3 Atomic Force Microscopy (AFM)*

152 Coating morphology was imaged using atomic force microscopy (AFM) in intermittent contact
153 mode (ICAFM) using an Autoprobe CP-Thermomicroscope. A sharp conical tip with a radius of
154 curvature <10 nm and an amplitude of vibration of 80 kHz (dLever Series Probe) mounted of a p-
155 type doped Si cantilever was used. Atomic force microscopy (AFM) was also used to determine the
156 PAA/CS-Ga thickness. Specifically, PAA/CS-Ga coating was deposited on half of a titanium sheet
157 and by scanning the sample from the PAA/CS-Ga on Ti to the Ti sheet and/or vice versa in different
158 points and using large scan area (90 μm x 90 μm), we measured the step between them. The
159 PAA/CS-Ga thickness was estimated to be $0.6 \pm 0.1 \mu\text{m}$.

160

161 *2.4.4 Three-point bending test*

162 The effect of PAA monolayer and PAA/CS-Ga bilayer on the mechanical performance of titanium
163 was assessed by means of three-point bending tests. The specimens (length: 50 mm, width: 10 mm,
164 thickness: 0.5 mm) were tested at a rate of 1 mm/min up to a strain limit of 0.03 mm/mm (3%)
165 using an INSTRON 5566 testing system (INSTRON, Buckinghamshire, UK). The support span-to-
166 depth ratio was 32 to 1. All the tests were carried out in physiological solution at $37.0 \pm 0.5^\circ\text{C}$.
167 Stress and strain (σ and ε , respectively), were evaluated as follows (Equation 1 and 2, respectively):

168
$$\sigma = \frac{3FL}{2bd^2} \quad \text{Equation 1}$$

169

170
$$\varepsilon = \frac{6Dd}{L^2} \quad \text{Equation 2}$$

171

172 where D is the deflection of the specimen at the middle of the support span, F is the load at a given
173 point of the load-deflection curve, L is the support span, b and d are the sample width and depth,
174 respectively.

175

176 **2.5 Bacterial strains, culture conditions and antimicrobial in vitro assays**

177 The antimicrobial activity of Ga(NO₃)₃ and PAA/CS-Ga coatings was evaluated against
178 *Escherichia coli* O157:H7 (avirulent strain) AD615 and *Pseudomonas aeruginosa* DSM 939. The
179 strain DSM 939 was purchased from DSMZ (Braunschweig, Germany) whereas the *E. coli* AD615
180 was gently provided by ADRIA Développement (Quimper Cedex, France). Both microorganisms
181 were grown as previously described (de Candia, Morea, & Baruzzi, 2015). Details on the
182 antimicrobial *in vitro* assays are reported in the Supplementary data (S1.2 Paragraph).

183

184 **2.6 Biological Characterization**

185 MG63 human osteoblast-like cells were used for biological characterization of the investigated
186 coatings. Experimental details on cell cultures, MMT viability assay, SEM-EDX analysis,
187 Fluorescence Microscopy, Real-time PCR (RT-qPCR) assay, mRNA expression quantification and
188 Western Blot analysis were reported in the Supplementary data (S1.3 Paragraph).

189

190 **2.7 Statistical analysis**

191 As far as the antimicrobial test are concerned, the significance of the change in microbial population
192 of each target strains at the beginning of incubation, in PBS at different Ga(NO₃)₃ concentrations as
193 well as when target strains were loaded on titanium sheets was calculated with the Friedman test,
194 (Ferguson, 1976). The null hypothesis defined as the absence of effect of the independent variable
195 (incubation conditions) on the dependent variable (viable cell count) was rejected at the level of
196 5%; then, the nonparametric *post hoc* Mann-Whitney test was applied in order to evaluate statistical
197 differences among microbial populations.

198 As far as the biological characterization is concerned, results were reported as mean ± standard
199 deviation of three different experiments. Gene expression differences were assessed by one-way
200 ANOVA and Bonferroni's T tests. Statistical significance was tested at p<0.05.

201 GraphPad Prism 4 (GraphPad Software, Inc., CA USA) was used to carry out statistical analysis.

202

203 **3. Results and discussion**

204 **3.1 Bilayer electrosynthesis**

205 Electrochemical polymerization of acrylic acid was carried out by cyclic voltammetry optimizing a
206 previously reported procedure (De Giglio et al., 2007) leading to a thin, strongly adherent PAA
207 coating on titanium substrate. A chronoamperometric deposition of CS in the presence of $\text{Ga}(\text{NO}_3)_3$
208 was achieved on a PAA-coated titanium sheet, obtaining a gallium-modified CS upper layer. This
209 strategy was followed since the CS coating failed to adhere directly on bare titanium. Indeed, the
210 presence of the PAA carboxyl groups has been exploited to promote the adhesion of the CS layer on
211 the metallic substrate. On the other hand, the chronoamperometric deposition of gallium ions, in
212 absence of CS, on a previously electrosynthesised PAA layer, produced a coating unsuitable to
213 release gallium ions due to the strong electrostatic interactions between Ga^{3+} and polymer's
214 carboxylate functionalities. Therefore, CS plays a pivotal role interacting with gallium ions, without
215 preventing their release. CS layers, both in presence and absence of $\text{Ga}(\text{NO}_3)_3$, were deposited onto
216 PAA-coated titanium electrodes by chronoamperometry for three electrodeposition times (15, 30
217 and 60 minutes). A current decrease was detected during the chronoamperometric step, thus
218 evidencing the CS or CS-Ga deposition on the underlying PAA coating (data not shown).

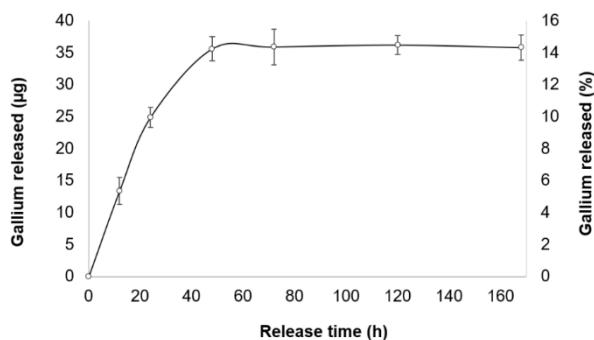
219

220 **3.2 Gallium quantification and release**

221 The total gallium amount entrapped in the coatings was estimated by DPASV after an acid
222 digestion. Different experiments, carried out varying the time of the chronoamperometric step,
223 showed that the gallium amount increased with the electrodeposition time. In particular,
224 electrodeposition time equal to 15, 30 and 60 minutes led to total gallium contents of $91 \pm 9 \mu\text{g}$, 242
225 $\pm 3 \mu\text{g}$ and $611 \pm 2 \mu\text{g}$, respectively. Therefore, it was possible to modulate the gallium amount in
226 the bilayer, tuning properly the electrodeposition time.

227 The coatings derived from the three different electrodeposition times showed the following features:
228 the 15 minutes-electrodeposited bilayers showed weak antimicrobial activity, while the 60 minutes-
229 electrodeposited bilayers displayed a cytotoxic behaviour (data not shown). Therefore, our attention
230 was focused only on the 30 minutes-electrodeposited coating since it combined antimicrobial and
231 cell-friendly performances.

232 In Figure 1, gallium release in PBS over time from the latter coating was reported. The release was
233 characterized by an initial burst, occurred within 24 hours, probably associated to the gallium ions
234 located within the surface CS layer of the specimen. When a plateau was reached, after 48 hours,
235 about 36 μg of gallium ions were released. After 7 days, only the 14.7% of the gallium amount
236 loaded into the coating was released. Indeed, it can be hypothesized that gallium ions, strongly
237 interacting with COO^- groups of the PAA matrix, were probably hindered to be released in the
238 investigated experimental conditions. On the other hand, the amount of gallium released, just after
239 24 hours, displayed proper antimicrobial performances (see “*In vitro* antibacterial activity”
240 Section).



241

242 **Figure 1:** Gallium release from PAA/CS-Ga bilayer over seven days in PBS solution at 37°C.

243

244 **3.3 Physico-chemical, morphological and mechanical characterization**

245 *3.3.1 Fourier Transform Infrared Spectroscopy in Attenuated Total Reflection mode (FT-IR/ATR)*

246 The FT-IR (ATR) spectra of pure PAA coating electropolymerized on Ti and pure CS film drop-
247 casted on Ti, as well as PAA/CS and PAA/CS-Ga bilayers, electrodeposited on Ti as previously
248 described, were shown in Figure 2.

249 In the PAA spectrum, a broad band due to the stretching of the –OH groups, falling in the range
250 3100-3600 cm^{-1} , was assigned to the O-H stretching of the hydrogen bonds between OH of COOH
251 groups and/or between adsorbed water molecules. The absorption band at 1703 cm^{-1} was assigned
252 to the C=O stretching vibration from carboxylic groups. Moreover, due to the dissociation of
253 COOH in COO^- groups, two stretching vibrations, $\nu(\text{C=O})_{\text{as}}$ and $\nu(\text{C=O})_{\text{s}}$ (asymmetric and
254 symmetric), were observed at 1550 and 1406 cm^{-1} , respectively.

255 The CS spectrum showed a broad band with two maximum at 3351 and 3290 cm^{-1} assigned to the
256 stretching vibration of O–H and N–H groups associated by intra and intermolecular hydrogen
257 bonding. At 2920 and 2871 cm^{-1} methylene and methyl C-H stretching were detected. Absorption at
258 about 1632 cm^{-1} (amidic C=O stretching) and 1520 cm^{-1} (N–H stretching) were characteristic of
259 amide groups. At 1419 and 1375 cm^{-1} bending vibration of methylene and methyl groups were
260 observed. Finally, absorptions at 1150, 1062 and 1024 cm^{-1} were assigned to CS's saccharide
261 structure (de Vasconcelos et al., 2006).

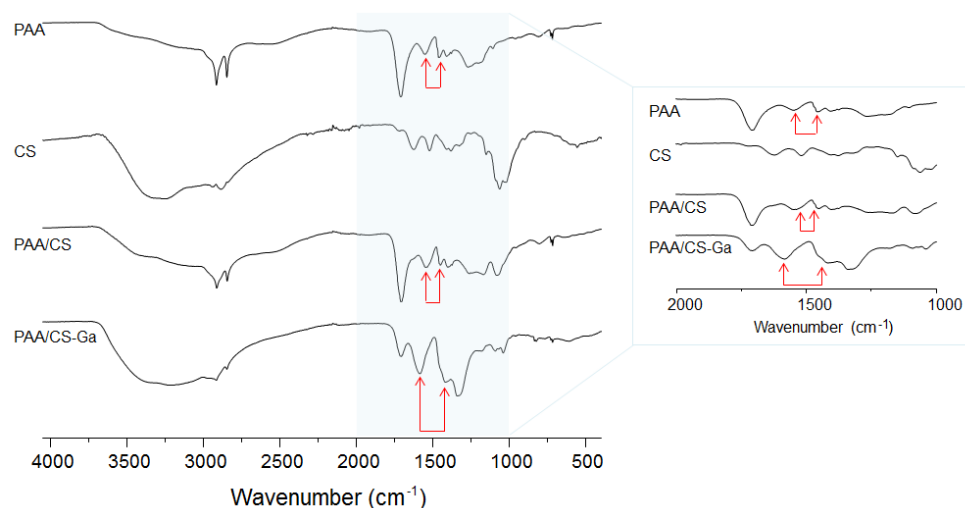
262 PAA/CS bilayer spectrum showed absorptions typical of PAA, in addition to absorptions in the
263 range 1180-1020 cm^{-1} , ascribable to the CS structure.

264 With regard to PAA/CS-Ga bilayer, the $\text{Ga}(\text{NO}_3)_3$ absorption bands were still evident (in particular,
265 the strong absorption band falling at 1323 cm^{-1} relevant to N-O stretching vibration).

266 An important feature was that the $\nu(\text{C=O})_{\text{as}}$ in PAA/CS bilayer fell at 1548 cm^{-1} , while in PAA/CS-
267 Ga fell at 1584 cm^{-1} , as well as the $\nu(\text{C=O})_{\text{s}}$ in PAA/CS bilayer fell at 1405 cm^{-1} , while in PAA/CS-
268 Ga fell at 1421 cm^{-1} .

269 The separation between these two $\nu(\text{C=O})$ was higher in the PAA/CS-Ga than in PAA/CS spectrum
270 (i.e., 163 vs. 143 cm^{-1}). In PAA spectrum, this separation was found equal to 144 cm^{-1} (Figure 2).

271 The higher band separation in PAA/CS-Ga supported the presence of a coordination or a cross-
272 linking between the COO⁻ groups of PAA and Ga³⁺ ions, as reported in a previous work on Ga-
273 citrate complexes (Clausen, Öhman, & Persson, 2005). This strong polymer-gallium interaction
274 could explain, as previously argued, the incomplete gallium release evidenced by the DPASV
275 experiments.

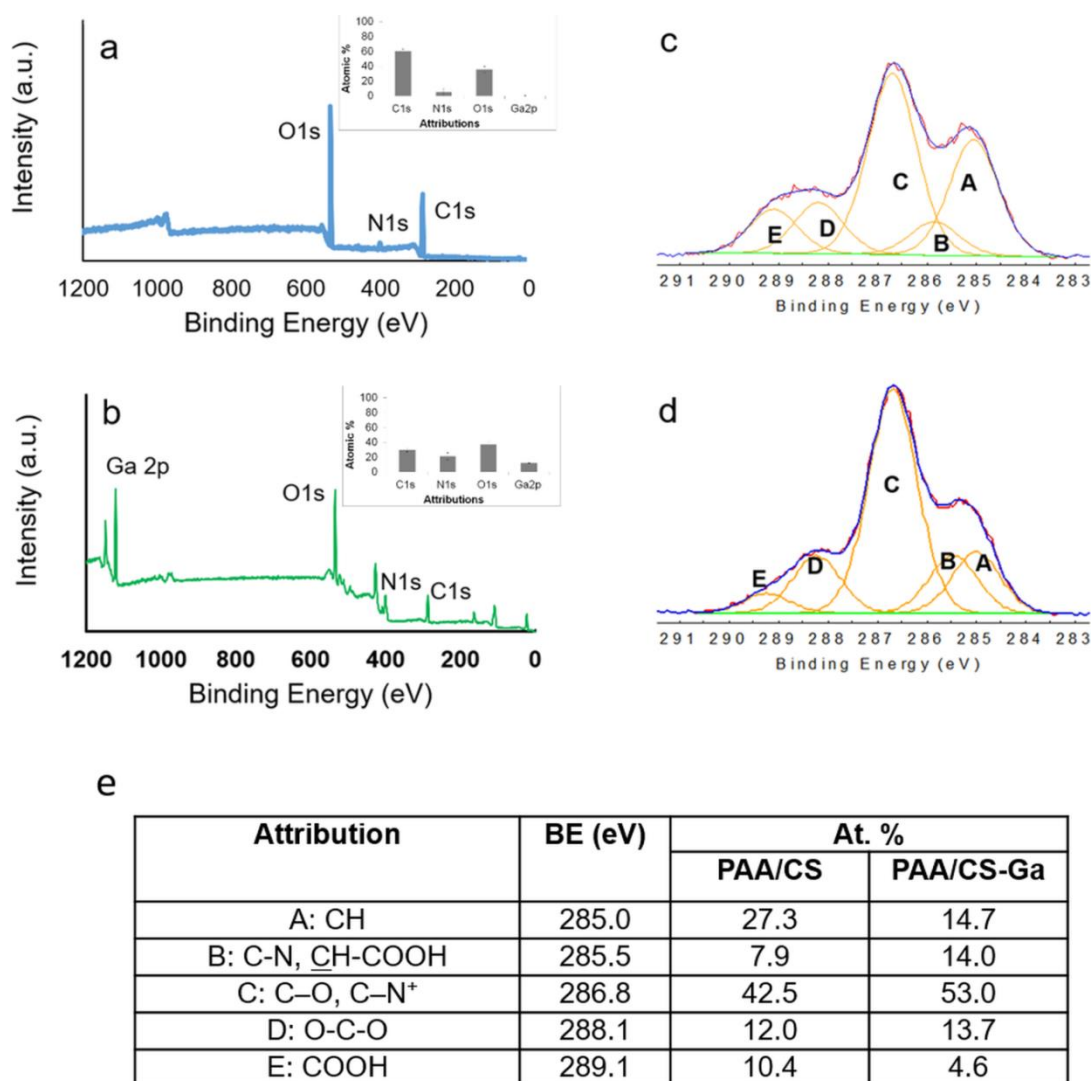


276
277 **Figure 2:** FT-IR (ATR) spectra relevant to PAA, CS (drop-casted on titanium), PAA/CS and
278 PAA/CS-Ga bilayers.

279 280 3.3.2 X-ray Photoelectron Spectroscopy (XPS)

281 XPS investigations were carried out both on PAA/CS bilayer and PAA/CS-Ga coatings, in order to
282 gain information about their surface chemical composition. In PAA/CS bilayer (Figure 3a), signals
283 relevant to oxygen, carbon and nitrogen were detected. In PAA/CS-Ga, a significant gallium Ga2p
284 main signal as well its minor signals were clearly detectable in the wide scan spectrum (Figure 3b).
285 The experimental BEs associated to Ga2p (Ga2p_{3/2}= 1118.7 eV) were ascribable to gallium (III)
286 compounds (Cossu et al., 1992). The C1s high-resolution spectrum of PAA/CS and PAA/CS-Ga
287 were also reported in Figure 3c and d, respectively: the relevant attributions and atomic percentages
288 were shown in Figure 3e. Curve fitting of C1s spectra revealed the typical functional groups of both

289 PAA and CS. Considering that peak D is relevant only to the CS moieties, it is possible to estimate
 290 the surface contribution of CS in the C1s signals of the bilayer with and without gallium. Indeed,
 291 the (peak D/peak E) ratio in pure CS was estimated to be 6.3, as already reported (Trapani et al.,
 292 2011). The same ratio in the PAA/CS bilayer resulted equal to 1.2, while in the PAA/CS-Ga bilayer
 293 this ratio was 3.0. These experimental results suggested that, when gallium was added during the
 294 second electrodeposition step, the presence of CS on the surface was increased.



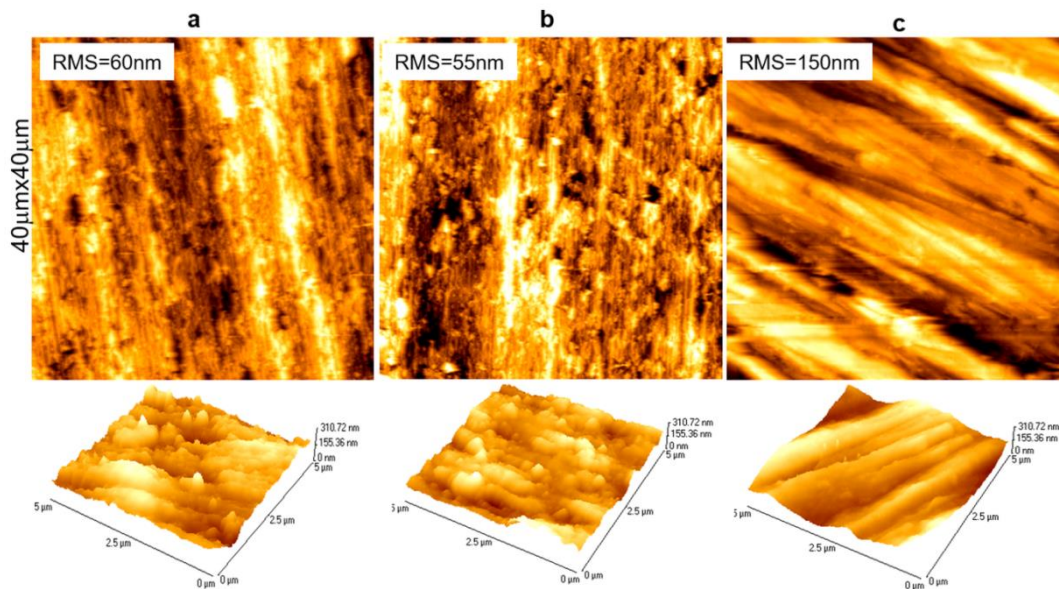
295
 296 **Figure 3:** XPS survey spectra relevant to PAA/CS (a) and PAA/CS-Ga bilayers on Ti (b). In the
 297 insets, the atomic percentages of the elements detected on the relevant bilayer surfaces are reported.
 298 C1s high-resolution curve fitting of PAA/CS (c) and PAA/CS-Ga bilayers (d), whose peak

299 attributions, (e) binding energies (BE) and atomic percentages (At. %) relevant to the C1s curve
300 fitting shown in panels (c) and (d). The maximum error on the peak position was ± 0.2 eV.

301

302 3.3.3 Atomic Force Microscopy (AFM)

303 Figure 4 compares $40\ \mu\text{m} \times 40\ \mu\text{m}$ topographical images of (a) Ti sheet, (b) PAA on Ti and (c)
304 PAA/CS-Ga bilayer on Ti. The morphology of the Ti sheet was characterized by parallel features,
305 i.e., grooves and ridges, which can be associated with abrading scratches (Qu et al., 2014; Peláez-
306 Abellán et al., 2012). When PAA was electrosynthesized on Ti, the film texture and surface
307 roughness did not change significantly.



308 **Figure 4:** AFM characterization. $40\ \mu\text{m} \times 40\ \mu\text{m}$ topographical images of (a) Ti electrode, (b) PAA on
309 Ti and (c) PAA/CS-Ga on Ti. Corresponding 3D $5\ \mu\text{m} \times 5\ \mu\text{m}$ AFM images are also compared at the
310 bottom with the same z-scale 0-310.72 nm.

311

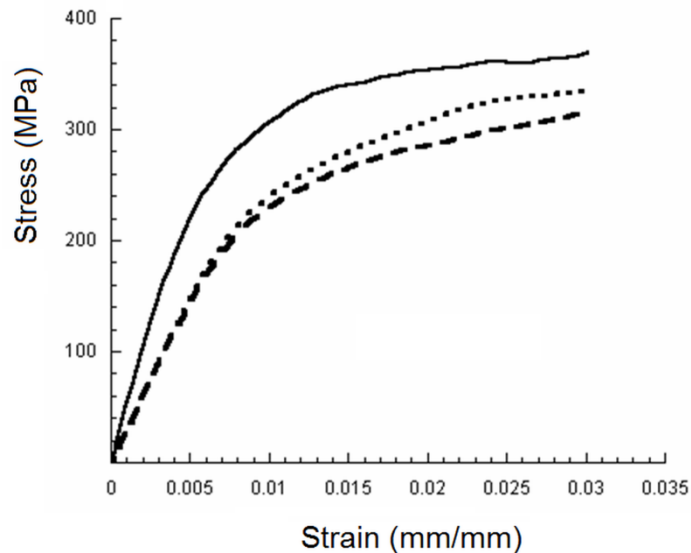
312 The fingerprint of the Ti crack structures and submicron heterogeneously-distributed morphology
313 near crack sites were clearly observed on the PAA film. On the contrary, when CS-Ga was
314 electrochemically deposited on PAA (obtaining PAA/CS-Ga bilayer), the morphology strongly
315 changed, resulting in well-oriented bundle formation (Dubey & He, 2012) and in increased

316 roughness. The bundle formation relevant to PAA/CS-Ga was highlighted by the corresponding 5
317 $\mu\text{m} \times 5 \mu\text{m}$ 3D images shown at the bottom of Figure 4.

318

319 3.3.4 Three-point bending test

320 Results from three-point bending tests on the different kinds of specimens (Ti, Ti/PAA monolayer,
321 Ti/PAA/CS-Ga bilayer) evidenced similar stress-strain curves (Figure 5). An initial linear region of
322 the stress–strain curve was evident. Then, a decrease of the slope was observed up to the strain limit
323 (0.03 mm/mm). Bending modulus (i.e., the slope of the linear region of the curve) and maximum
324 stress were evaluated as mean value \pm standard deviation.



325

326 **Figure 5:** Typical stress-strain curves obtained from three-point bending tests on the different kinds
327 of devices: Ti (full line), Ti/PAA monolayer (dotted line), Ti/PAA/CS-Ga bilayer (dashed line).

328

329 Ti samples provided higher values of modulus (58 ± 5 GPa) and maximum stress (370 ± 30 MPa)
330 than those obtained from Ti/PAA monolayer (41 ± 5 GPa and 340 ± 30 MPa) and Ti/PAA/CS-Ga
331 bilayer (40 ± 5 GPa and 310 ± 20 MPa).

332 It is worth noting that, in the three-point bending test, stress reached the maximum value in the

333 outer regions of the specimen. As a consequence of the polymeric layer, Ti/PAA system provided

334 values of modulus lower than those obtained from Ti. Even though the support span-to-depth ratio
335 was 32 to 1, taking into account that a shear stress is induced as the bending moment increases
336 linearly from zero at the supports to a maximum value at the central loading point, the three-point
337 bending test may also provide an information on the metal-polymer interface.

338 In comparison to Ti samples, the results evidenced that the surface coatings did not dramatically
339 alter the maximum flexural stress achieved at 0.03 mm/mm; in addition, the mechanical stability of
340 the coatings (i.e., Ti/PAA monolayer and Ti/PAA/CS-Ga bilayer) can be deduced.

341

342 **3.4 In vitro antibacterial activity**

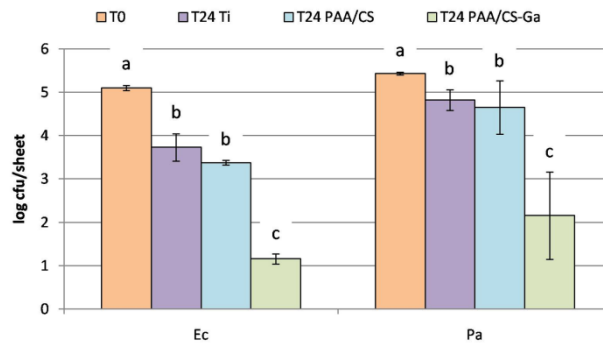
343 It is generally accepted that the main biological mechanism of antimicrobial activity of Ga(III) is
344 based on the competition with Fe(III): the gallium-siderophore complexes deregulate several
345 microbial iron-based metabolisms causing cell death (Kelson, Carnevali & Truong-Le, 2013).
346 Differently from Kaneko et al. (2007) who evaluated microbial growth kinetic in 1% TSB
347 supplemented with several Ga(NO₃)₃ concentrations, here we assayed antimicrobial activity of
348 Ga(NO₃)₃ against two microbial populations in a PBS lacking of carbon, nitrogen sources and other
349 nutrients. As concerns *P. aeruginosa*, Kaneko et al. (2007) demonstrated that strain PA01,
350 inoculated in 1% TSB, was proportionally killed by increasing Ga(NO₃)₃ concentration up to 1 mM
351 when it was added during PA01 stationary growth phase. In accordance with this result, our study
352 showed the best killing activity against *P. aeruginosa* DSM 939 (under the stationary growth phase
353 in PBS), increasing Ga(NO₃)₃ up to 10 mM (see section S2.1 and Fig. S2).

354 The antimicrobial activity of the coatings was verified by applying an agar plate method derived
355 from the ASTM E2180 (Standard Test Method for Determining the Activity of Incorporated
356 Antimicrobial Agent(s) in Polymeric or Hydrophilic Materials) as already carried out for zinc
357 oxalate nanoparticles copolymerized with acrylates (Sontakke et al., 2012) and cobalt ferrite
358 nanoparticles (Sanpo et al., 2013).

359 The application of Friedmann test to viability values of *E. coli* AD615 and *P. aeruginosa* DSM 939
360 on titanium sheet (before and after incubation), PAA/CS and PAA/CS-Ga coatings, showed that
361 these data did not belong to the same population of values. Then, a pairwise comparison of
362 microbial values from PAA/CS versus PAA/CS-Ga coatings was carried out by applying the Mann-
363 Whitney test. As shown in Figure 6, viability of both target strains decreased (about two log
364 cfu/sheet) significantly in comparison of initial inoculum level (T0) although no differences were
365 found in viable cell count on titanium sheets and PAA/CS coatings. At contrary, viable cell counts
366 of both target strains was reduced significantly ($p < 0.05$) when they were loaded on titanium sheets
367 on which $\text{Ga}(\text{NO}_3)_3$ was electrodeposited (PAA/CS-Ga coatings). Furthermore, the LIVE/DEAD®
368 BacLight™ assay provided a qualitative evidence of PAA/CS-Ga coatings effectiveness (see
369 Paragraph S2.2 and Fig.S3).

370 In comparison with $\text{Ga}(\text{NO}_3)_3$ antimicrobial assays, trials with PAA/CS-Ga coatings produced a
371 better result. In absence of direct evidences, it is possible to hypothesize that PAA/CS-Ga coatings
372 released a higher amount of Ga(III) than that occurring in $\text{Ga}(\text{NO}_3)_3$ solutions, due to the
373 precipitation of large part of Ga(III) as $\text{Ga}(\text{OH})_3$ when the $\text{Ga}(\text{NO}_3)_3$ was dispersed in buffer with
374 pH range between pH 3 and 9 (Rzhepishavska et al., 2011). On the other hand, an additional
375 antimicrobial effect of gallium retained in the coatings could not be excluded. Moreover, the
376 reduced mobility of microbial cells into the agar matrix, as happens in microbial biofilms, could be
377 a further reason of the increased antimicrobial efficacy of gallium ions, that were demonstrated to
378 inhibit biofilm formation in *in vitro* assays (Kaneko et al., 2007).

379



380

381 **Figure 6:** Antimicrobial activity. Viable cell count of *Escherichia coli* O157:H7 AD615 (Ec) and
 382 *Pseudomonas aeruginosa* DSM 939 (Pa) before incubation (T0) and after 24 h at 37°C loaded onto
 383 titanium sheets (T24 Ti), PAA/CS (T24 PAA/CS) and PAA/CS-Ga bilayers (T24 PAA/CS-Ga).

384 The data were consolidated from three independent experiments; bars represent mean values \pm
 385 standard deviation. The same letter represents values statistically similar ($p > 0.05$) within the same
 386 strain.

387

388 **3.5 Biological evaluation**

389 Considering the role of osteoblastic cells in the bone remodelling process, we investigated whether
 390 the PAA/CS-Ga coating could affect osteoblastic behaviour in terms of viability, proliferation and
 391 relevant osteoblastic markers. We used the osteoblastic cell line MG63 (Billiau et al., 1977), which
 392 is a well-known osteosarcoma cell line commonly used as osteoblastic model to test titanium
 393 implants (Martin et al., 1995; De Giglio et al., 2010). Tests were performed at two different time
 394 points: 2 days (T1) was chosen to evaluate the effect of the burst release of gallium, whilst 7 days
 395 (T2) was selected to test its effect at a longer time.

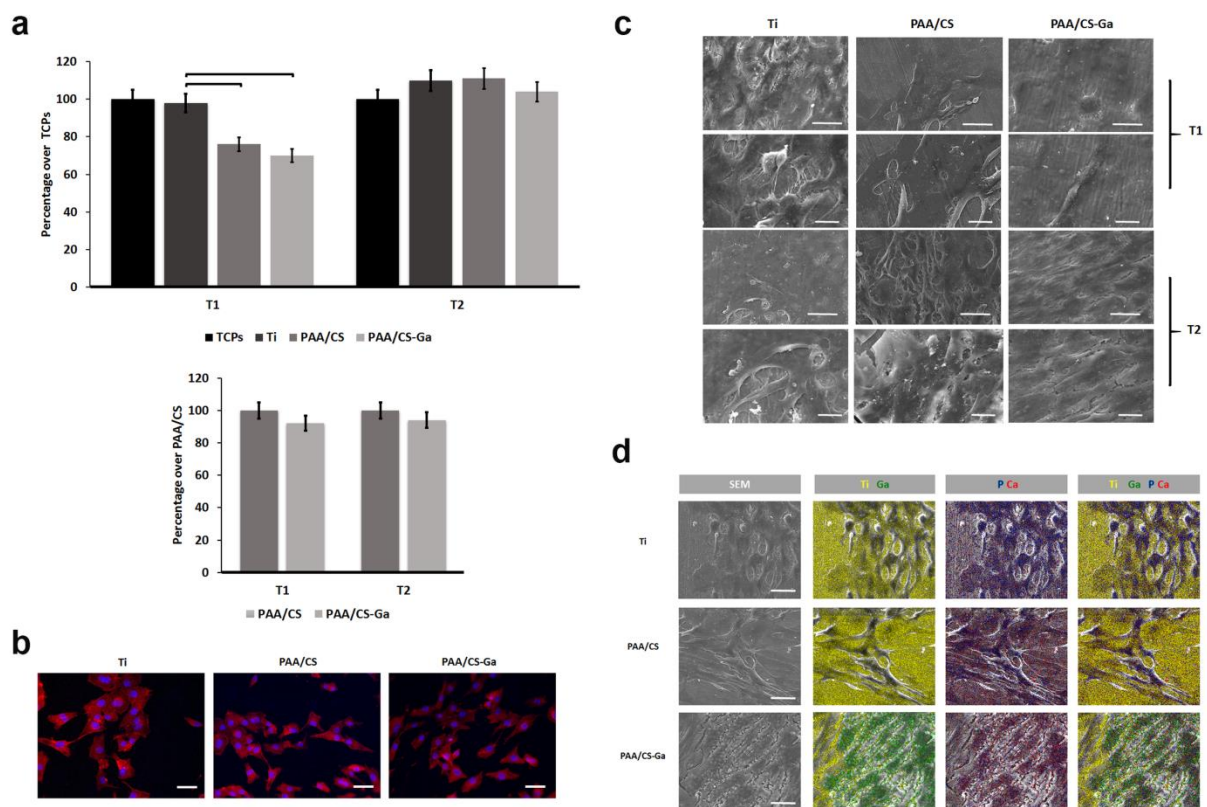
396 Viability on PAA/CS and PAA/CS-Ga coatings was investigated by MTT assay: no significant
 397 differences were detected between MG63 cultured on tissue culture plates (TCPs) and those seeded
 398 on bare Ti at both time points analysed (Figure 7a). At T1 a significant reduction in cell viability in
 399 comparison with controls was detected for both Ti coatings (i.e., PAA/CS and PAA/CS-Ga), even if
 400 with values higher than 65%. This reduction could be at least in part due to a different

401 adhesion/proliferation of cells between chitosan and plastic surfaces. Comparing absorbance data
402 obtained from cells cultured on the two different Ti coatings, it can be argued that the observed
403 reduction in cell viability was not ascribable to gallium doping. No significant differences were
404 detectable in all samples at T2.

405 Cytoskeletal detection was performed at T1 in order to ascertain the initial cell adhesion to the
406 substrates, since this event is crucial in implant osseointegration (Mattioli-Belmonte et al., 2014).

407 Actin fluorescence images of MG63 (Figure 7b) showed well spread cells on all analysed samples,
408 with a preserved organization of the cytoskeletal structures that determines the global cell shape.

409 SEM observations followed the same trend of MTT results: at T1, a higher cell colonization was
410 detected on bare Ti in comparison with the two tested coatings, whilst at T2 cells distribution was
411 almost homogeneous onto all analysed surfaces, with a typical spindle-shaped morphology (Pautke
412 et al., 2004) (Figure 7c). Interestingly, cell orientation on PAA/CS-Ga coating was superimposable
413 to surface bundles detected with AFM, suggesting that the changes in topography could represent
414 an instructive pattern for cell arrangement. Moreover, EDX mapping (Figure 7d) evidenced the
415 distribution of Ti, Ca, P and Ga on the different cultured surfaces and confirmed that the presence
416 of gallium did not affect cell adhesion and spreading.



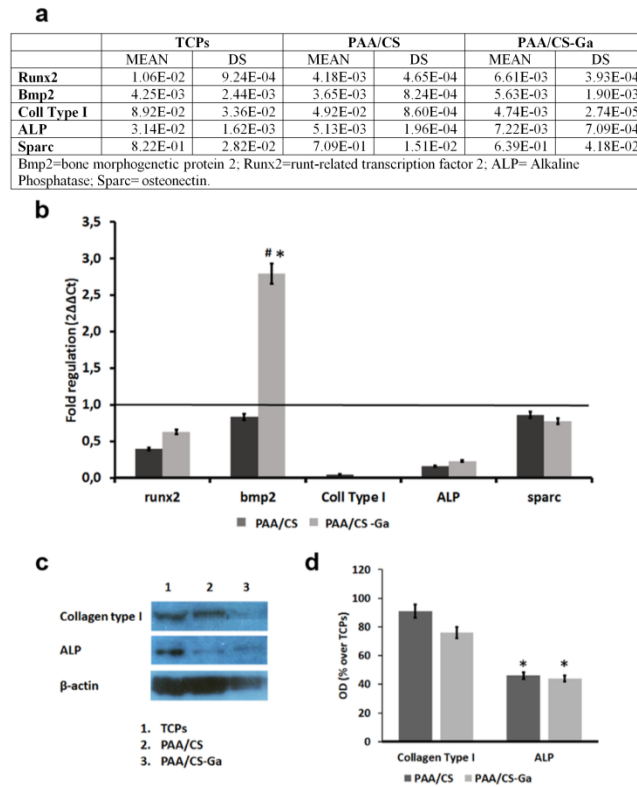
417 **Figure 7:** Cell adhesion and viability: a) MTT viability test of MG63 cultured for 2 (T1) and 7
 418 (T2) days on Tissue Culture plates (TCPs), bare Titanium (Ti), PAA/CS and PAA/CS-Ga bilayers.
 419 Data are expressed as percentage of TCPs control culture or of PAA/CS, Mean values \pm SD are
 420 reported; brackets indicate statistic significant ($p < 0.05$) differences; b) Fluorescence microscopy
 421 images of actin filament (red) distribution in cell cultured on Ti, PAA/CS and PAA/CS-Ga, nuclei
 422 are stained in blue (scale bars 10 μm); c) Scanning electron micrographs of cells (asterisks) cultured
 423 on Ti, PAA/CS and PAA/CS-Ga at both time point analysed (scale bars 20 μm); d) EDX mapping
 424 of element distribution (Ti=yellow; Ga=green; P=blue, Ca=red) in cell cultures at T2 (scale bars 20
 425 μm).

426

427 To evaluate whether the proposed coatings (i.e. PAA/CS and PAA/CS-Ga bilayers) may affect
 428 MG63 behaviour, mRNA expression of genes involved in bone growth was evaluated. Analysis of
 429 their relative expression ($2^{-\Delta\text{ct}}$) evidenced a reduction in mRNA between cells seeded on TCPs and

430 those seeded on the two coatings; these differences were significantly marked for runx2, collagen
 431 type I and ALP (Figure 8a).

432 In order to better elucidate if these changes were attributable to gallium doping, we analysed mRNA
 433 fold regulation ($2^{-\Delta\Delta Ct}$) in comparison to control cultures (TCPs).



434

435 **Figure 8:** MG63 gene and protein expression: a) mRNA relative expression; b) Histogram depict
 436 changes in mRNA expression in MG63 cultured for 7 days (T2) days on PAA/CS and PAA/CS-Ga
 437 bilayers. Data are expressed as Fold-regulation ($2^{-\Delta\Delta Ct}$) (see S1.3 Paragraph). The black line
 438 indicates the range of physiological mRNA expression changes; * $p < 0.05$ vs TCPs; # $p < 0.05$ vs
 439 PAA/CS; c) Western blot analysis of Collagen Type I and Alkaline Phosphatase (ALP) protein
 440 content in MG63 cultured on TCPs, PAA/CS and PAA/CS-Ga bilayers; d) Densitometric
 441 quantitation of the blots presented in c. Results are expressed as percentage of control cells (TCPs);
 442 * $p < 0.05$ vs TCPs.

443

444 This analysis suggested that gallium did not produce negative effect on osteoblastic marker
445 expression, on the contrary it appeared to enhance (2.5 fold) bmp2 mRNA expression (Figure 8b).
446 As far as Collagen Type I and ALP are concerned, qRT-PCR showed a significant reduction in their
447 mRNA expression in comparison with the controls; at contrary, coatings unaffected osteonectin
448 (sparc) expression. Western Blot analysis detected an unchanged expression of Collagen Type I
449 protein content between the coatings and control cultures and a significant reduction of ALP on
450 both the tested coatings respect to TCPs (Figures 8c and d). However, no significant differences
451 related to the presence of gallium were observed. The detected decrease in ALP production on the
452 gallium-modified coating is in contrast with recent literature (Cochis et al., 2016; Verron et al.,
453 2010). This evidence could be, at least in part, related to the different pattern of expression of
454 MG63 in comparison with other human (i.e. Saos2) or murine (i.e. MC3T3E-1) osteoblastic cell
455 lines at the analysed time point (Pautke et al., 2004). On the contrary, the positive effect of gallium
456 is appreciable in our system by the detection of the upregulation of bmp2, which is a genetic marker
457 of the early osteoblastic differentiation. Overall, biological data are consistent with the lack of
458 adverse effects of the proposed coatings up to 7 days.

459

460 **4. Conclusions**

461 A novel gallium-modified coating on titanium implants has been proposed to the aim of preventing
462 microbial colonization without affecting osseointegration. The investigated PAA/CS-Ga bilayer was
463 obtained on titanium substrates by low-cost electrochemical techniques. Since gallium exerts its
464 antibacterial activity by “mimicking” Fe(III), the application of PAA/CS-Ga coatings releasing
465 Ga(III) ions could produce an antimicrobial effect, in particular when microbial cells start to adhere
466 to inert surfaces, as happens during first stage of biofilm formation. In addition, viability and
467 proliferation of MG63 osteoblast-like cells were unaffected by the presence of gallium in the
468 PAA/CS-Ga bilayer: among several genetic marker evaluated, gallium upregulated bmp2, a genetic
469 marker of early osteoblastic differentiation. Biological data of PAA/CS-Ga bilayer on osteoblast-

470 like cell lines and bacterial pathogens strengthen its effectiveness for a potential application of this
471 coating in orthopaedic and/or dental field.

472

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477

478 **References**

479 Berlutti, F., Superti, F., Nicoletti, M., Morea, C., Frioni, A., Ammendolia, M. G., Battistoni
480 A., Valenti, P. (2008) Bovine lactoferrin inhibits the efficiency of invasion of respiratory A549 cells
481 of different iron-regulated morphological forms of *Pseudomonas aeruginosa* and *Burkholderia*
482 *cenocepacia*. *International journal of immunopathology and pharmacology*, 21(1), 51-59.

483 Bernstein, L. R., (1998) Mechanisms of therapeutic activity for gallium. *Pharmacological*
484 *reviews*, 50(4), 665-682.

485 Bernstein, L. R. (2013) Gallium, therapeutic effects. In Kretsinger, R.H.; Uversky, V.N.;
486 Permyakov, E.A, *Encyclopedia of Metalloproteins* (pp. 823-835) Springer New York.

487 Billiau, A., Edy, V. G., Heremans, H., Van Damme, J., Desmyter, J., Georgiades, J. A., &
488 De Somer, P. (1977). Human interferon: mass production in a newly established cell line, MG-63.
489 *Antimicrobial agents and chemotherapy*, 12(1), 11-15.

490 Clausén, M., Öhman, L. O., & Persson, P. (2005). Spectroscopic studies of aqueous gallium
491 (III) and aluminum (III) citrate complexes. *Journal of inorganic biochemistry*, 99(3), 716-726.

492 Cochis, A., Azzimonti, B., Della Valle, C., De Giglio, E., Bloise, N., Visai, L., ... & Chiesa,
493 R. (2016). The effect of silver or gallium doped titanium against the multidrug resistant
494 *Acinetobacter baumannii*. *Biomaterials*, 80, 80-95.

495 Cossu, G., Ingo, G. M., Mattogno, G., Padeletti, G., & Proietti, G. M. (1992). XPS
496 investigation on vacuum thermal desorption of UV/ozone treated GaAs (100) surfaces. *Applied*
497 *surface science*, 56, 81-88.

498 Danielsson, L. G., Jagner, D., Josefson, M., Westerlund, S. (1981) Computerized
499 potentiometric stripping analysis for the determination of cadmium, lead, copper and zinc in
500 biological materials. *Analytica Chimica Acta*, 127, 147-156.

501 De Candia, S., Morea, M., & Baruzzi, F. (2015). Eradication of high viable loads of *Listeria*
502 *monocytogenes* contaminating food-contact surfaces. *Frontiers in microbiology*, 6, art. n. 733.

503 De Giglio, E., Cometa, S., Cioffi, N., Torsi, L., Sabbatini, L. (2007) Analytical
504 investigations of poly (acrylic acid) coatings electrodeposited on titanium-based implants: a
505 versatile approach to biocompatibility enhancement. *Analytical and bioanalytical chemistry*, 389(7-
506 8), 2055-2063.

507 De Giglio, E., Cometa, S., Ricci, M. A., Zizzi, A., Cafagna, D., Manzotti, S., ... & Mattioli-
508 Belmonte, M. (2010). Development and characterization of rhVEGF-loaded poly (HEMA–MOEP)
509 coatings electrosynthesized on titanium to enhance bone mineralization and angiogenesis. *Acta*
510 *biomaterialia*, 6(1), 282-290.

511 De Giglio, E., Cafagna, D., Cometa, S., Allegretta, A., Pedico, A., Giannossa, L. C.,
512 Sabbatini, L., Mattioli-Belmonte, M., Iatta, R. (2013) An innovative, easily fabricated, silver
513 nanoparticle-based titanium implant coating: development and analytical characterization.
514 *Analytical and bioanalytical chemistry*, 405(2-3), 805-816.

515 de Vasconcelos, C. L., Bezerril, D. P., Dos Santos, D. E. S., Dantas, D. T., Pereira, M. R., Fonseca,
516 J. L. C. (2006) Effect of molecular weight and ionic strength on the formation of polyelectrolyte
517 complexes based on poly (methacrylic acid) and chitosan. *Biomacromolecules*, 7(4), 1245-1252.

518 Ferguson G. A. (1976). *Statistical Analysis in Psychology and Education* (4th ed.). New
519 York: McGraw-Hill.

520 Hedrick, T. L., Adams, J. D., Sawyer, R. G. (2006) Implant-associated infections: an
521 overview. *Journal of long-term effects of medical implants*, 16(1) 83-99.

522 Dubey, M., & He, H. (2012). *Morphological and Photovoltaic Studies of TiO₂ NTs for High*
523 *Efficiency Solar Cells*. INTECH Open Access Publisher.

524 Kaneko, Y., Thoendel, M., Olakanmi, O., Britigan, B. E., & Singh, P. K. (2007). The
525 transition metal gallium disrupts *Pseudomonas aeruginosa* iron metabolism and has antimicrobial
526 and antibiofilm activity. *The Journal of clinical investigation*, 117(4), 877-888.

527 Kelson, A. B., Carnevali, M., & Truong-Le, V. (2013). Gallium-based anti-infectives:
528 targeting microbial iron-uptake mechanisms. *Current opinion in pharmacology*, 13(5), 707-716.

529 Martin, J. Y., Schwartz, Z., Hummert, T. W., Schraub, D. M., Simpson, J., Lankford, J., ...
530 & Boyan, B. D. (1995). Effect of titanium surface roughness on proliferation, differentiation, and
531 protein synthesis of human osteoblast-like cells (MG63). *Journal of biomedical materials research*,
532 29(3), 389-401.

533 Mattioli-Belmonte, M., Cometa, S., Ferretti, C., Iatta, R., Trapani, A., Ceci, E., ... & De
534 Giglio, E. (2014). Characterization and cytocompatibility of an antibiotic/chitosan/cyclodextrins
535 nanocoating on titanium implants. *Carbohydrate polymers*, 110, 173-182.

536 Muzzarelli, R. A. A. (2009) Chitins and chitosans for the repair of wounded skin, nerve,
537 cartilage and bone. *Carbohydrate polymers*, 76(2), 167-182.

538 Pautke, C., Schieker, M., Tischer, T., Kolk, A., Neth, P., Mutschler, W., & Milz, S. (2004).
539 Characterization of osteosarcoma cell lines MG-63, Saos-2 and U-2 OS in comparison to human
540 osteoblasts. *Anticancer research*, 24(6), 3743-3748.

541 Peláez-Abellán, E., Duarte, L. T., Biaggio, S. R., Rocha-Filho, R. C., & Bocchi, N. (2012).
542 Modification of the titanium oxide morphology and composition by a combined chemical-
543 electrochemical treatment on cp Ti. *Materials Research*, 15(1), 159-165.

544 Qu, Q., Wang, L., Chen, Y., Li, L., He, Y., & Ding, Z. (2014). Corrosion Behavior of
545 Titanium in Artificial Saliva by Lactic Acid. *Materials*, 7(8), 5528-5542.

546 Rack, H. J., & Qazi, J. I. (2006). Titanium alloys for biomedical applications. *Materials*
547 *Science and Engineering: C*, 26(8), 1269-1277.

548 Rzhapishevska, O., Ekstrand-Hammarström, B., Popp, M., Björn, E., Bucht, A., Sjöstedt, A.,
549 ... & Ramstedt, M. (2011). The antibacterial activity of Ga³⁺ is influenced by ligand complexation
550 as well as the bacterial carbon source. *Antimicrobial agents and chemotherapy*, 55(12), 5568-5580.

551 Sanpo, N., Berndt, C. C., Wen, C., & Wang, J. (2013). Transition metal-substituted cobalt
552 ferrite nanoparticles for biomedical applications. *Acta biomaterialia*, 9(3), 5830-5837.

553 Scofield, J. H. (1976) Hartree-Slater subshell photoionization cross-sections at 1254 and
554 1487 eV. *Journal of Electron Spectroscopy and Related Phenomena*, 8(2), 129-137.

555 Sidambe, A. T. (2014). Biocompatibility of advanced manufactured titanium implants—A review.
556 *Materials*, 7(12), 8168-8188.

557 Sontakke, T. K., Jagtap, R. N., Singh, A., & Kothari, D. C. (2012). Nano ZnO grafted on
558 MAA/BA/MMA copolymer: An additive for hygienic coating. *Progress in Organic Coatings*,
559 74(3), 582-588.

560 Trampuz, A., Osmon, D. R., Hanssen, A. D., Steckelberg, J. M., Patel, R. (2003) Molecular
561 and antibiofilm approaches to prosthetic joint infection. *Clinical orthopaedics and related research*,
562 414, 69-88.

563 Trapani, A., De Giglio, E., Cafagna, D., Denora, N., Agrimi, G., Cassano, T., Gaetani, S.,
564 Cuomo, V., Trapani, G. (2011) Characterization and evaluation of chitosan nanoparticles for
565 dopamine brain delivery. *International journal of pharmaceutics*, 419(1), 296-307.

566 Udisti, R., & Piccardi, G. (1988). Determination of gallium traces by differential pulse
567 anodic stripping voltammetry. *Fresenius' Zeitschrift für analytische Chemie*, 331(1), 35-38.

568 Varma, A. J., Deshpande, S. V., & Kennedy, J. F. (2004). Metal complexation by chitosan
569 and its derivatives: a review. *Carbohydrate Polymers*, 55(1), 77-93.

570 Verron, E., Masson, M., Khoshniat, S., Duplomb, L., Wittrant, Y., Baud'huin, M., Badran,
571 Z., Bujoli, B., Janvier, P., Scimeca, J. C., Bouler, J. M., Guicheux, J., (2010) Gallium modulates
572 osteoclastic bone resorption *in vitro* without affecting osteoblasts. *Br J Pharmacol.* 159(8):1681-92.